

# INVESTIGATION ON ELECTROCHEMICAL PROPERTIES OF ELECTROCHROMIC TUNGSTEN TRIOXIDE THIN FILMS

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## Abstract

Electrochemical properties of electrochromic (EC) tungsten trioxide (WO<sub>3</sub>) thin films on the fluorine doped tin oxide (FTO) coated glass substrates have been studied. Different molar concentrations of WO<sub>3</sub> precursor solution were sprayed on FTO glass substrates to synthesize amorphous and crystalline phase tungsten trioxide thin films by conducting different annealing temperature. From the different WO<sub>3</sub> precursor concentrations, 0.05 M enabled to increase current densities of WO<sub>3</sub> film. The present study indicated that the optimum processing condition of annealing temperature 400 °C and WO<sub>3</sub> precursor concentration 0.05 M would produce the WO<sub>3</sub> film with higher electrochemical current densities which is attributed to a higher crystallinity and lower band gap energy. In examining the electrochromic performance of WO<sub>3</sub> film, its color is blue upon coloring and colorless upon bleaching and the coloration efficiency is 1.89 cm<sup>2</sup>C<sup>-1</sup> in KI and 2.01 cm<sup>2</sup>C<sup>-1</sup> in KCl electrolytes. The color change would be associated with the intercalation (deintercalation) of K<sup>+</sup> ions. The diffusion of ions in the insertion/ extraction process thus play a role. The diffusion coefficient of WO<sub>3</sub> film is observed to be quite high and is in the range of 1.1 – 1.4 × 10<sup>-12</sup> cm<sup>2</sup>s<sup>-1</sup> in both KI and KCl electrolytes. The higher electrochemical activity renders it as promising material for smart windows and energy saving applications.

**Keywords:** Tungsten trioxide film, Electrochromic behaviors, Smart Window

## Introduction

Tungsten trioxide (WO<sub>3</sub>) is the most studied among the electrochromic materials and exhibits an n-type semiconducting behavior with a band gap energy of 2.5-3.3 eV. [ Jayachandran M. *et al.*]. Specifically, it is a cathodic electrochromic material used as a working electrode in electrochromic devices. It is a transition metal oxide with excellent chemical and thermal stability. It has promising electrochromic properties such as high coloration efficiency and fast switching response. It is used as a functional layer in the applications of gas sensors, solar cells and electrochromic devices such as smart windows can meet the market demand of energy-saving devices. There are various choices for preparing WO<sub>3</sub> films with the development of thin film technology. These include sputtering, chemical vapor deposition, spray pyrolysis, evaporation, sol-gel and laser ablation deposition. Among these, spray pyrolysis method is the most cost-effective for producing large-area films, and enables a better control over the film growth and hence more homogenous and more transparent films are yielded [M.F. Daniel *et al.*, (1987)].

## Materials and Methods

### Preparation of Tungsten trioxide (WO<sub>3</sub>) thin films

Tungsten trioxide (WO<sub>3</sub>) film was prepared using spray pyrolysis method to be used as electrochromic layer in the electrochromic device. Electrochromic devices (ECDs) are normally composed of five layers as shown in Figure 1. Tungsten trioxide (WO<sub>3</sub>) powder (0.03 M, 0.05 M and 0.07 M) was dissolved in 50 ml of ammonia at 80 °C under continuous stirring for 30 min to form tungsten trioxide (WO<sub>3</sub>) precursor solution. The obtained WO<sub>3</sub> solution was diluted with 10 ml of deionized water and then cooled down to room temperature and filtered. After that, the

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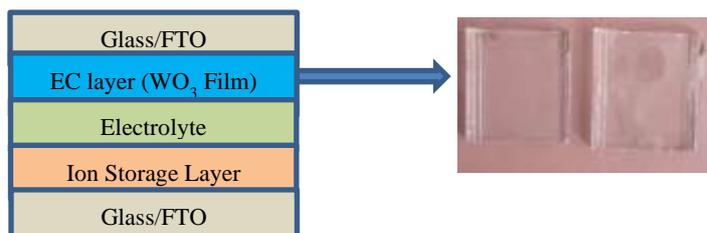
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final  $\text{WO}_3$  solution was sprayed on the FTO substrate (substrate temperature is  $300\text{ }^\circ\text{C}$ ) by using spray pyrolysis method to obtain  $\text{WO}_3$  films. The resulting  $\text{WO}_3$  films with the varying concentration of (0.03 M, 0.05 M and 0.07 M) were annealed at  $400\text{ }^\circ\text{C}$  for 1 h.



**Figure 1** Basic design of electrochromic device

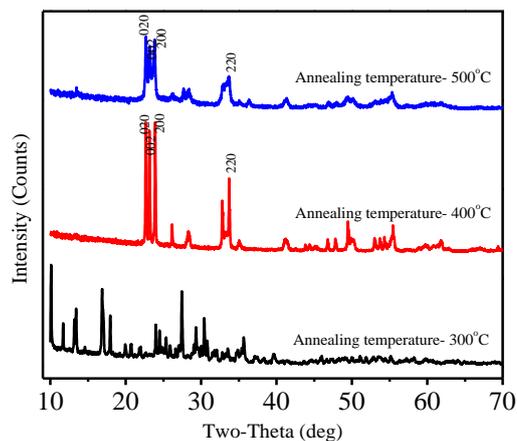
### Characterization of $\text{WO}_3$ Films

The structural properties of  $\text{WO}_3$  films were investigated X-ray diffractometer (RIGAKU Multiflex (Japan)). The optical properties of  $\text{WO}_3$  films were performed using UV-vis spectrophotometry (Genesys 10S). The electrochemical properties of  $\text{WO}_3$  films were characterized by cyclic voltammetry. (Electrochemical Workstation- CorrTest CS350). Taking the optimum  $\text{WO}_3$  films (annealing temperature  $400\text{ }^\circ\text{C}$ , 0.05 M) the coloration efficiency of  $\text{WO}_3$  films with different electrolyte (KI & KCl) were measured. Cyclic voltammetry was used to obtain the electrochemical properties of the  $\text{WO}_3$  films under oxidation state. During coloration the  $\text{WO}_3$  film becomes blue color and turns into colorless during the bleaching. The transmission of the samples at colored state and bleached state was measured by UV-vis spectrophotometry.

## Results and Discussion

### Crystalline Structure of $\text{WO}_3$ Films

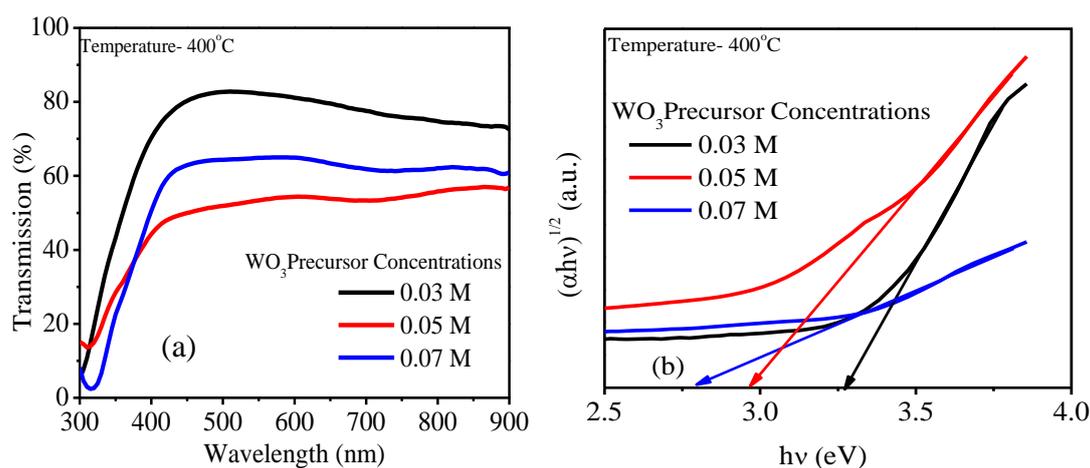
XRD profiles of  $\text{WO}_3$  films annealed at different temperatures are depicted in Figure 2. From the XRD measurement,  $\text{WO}_3$  films annealed at  $300\text{ }^\circ\text{C}$  shows amorphous nature. At the elevated temperatures  $400\text{ }^\circ\text{C}$  and  $500\text{ }^\circ\text{C}$ ,  $\text{WO}_3$  films reveal the orthorhombic structure. The main diffraction peak positions of  $\text{WO}_3$  films for  $400\text{ }^\circ\text{C}$  and  $500\text{ }^\circ\text{C}$  are observed at  $2\theta$  angle of  $22^\circ$ ,  $23^\circ$ ,  $24^\circ$  and  $34^\circ$  with the corresponding plane of (020), (002), (200) and (220). The main diffraction peak intensity of  $\text{WO}_3$  films annealed at  $400\text{ }^\circ\text{C}$  shows the stronger peak intensity. It may be due to the formation of better crystallinity. At the annealing temperature  $500\text{ }^\circ\text{C}$ , the main diffraction peak positions  $\text{WO}_3$  films are still the same but the width of the diffraction peak position and intensity are slightly broad and declined.



**Figure 2** XRD profiles of  $\text{WO}_3$  films with varying annealing temperatures

### Optical Transmission and Band Gap Energy of WO<sub>3</sub> Films

Figure 3 (a) shows the optical transmission spectra of WO<sub>3</sub> films for all WO<sub>3</sub> precursor concentrations. As can be seen in Figure 3 (a), the values of the optical transmission of WO<sub>3</sub> films decreased from 80% to 65% upon increasing WO<sub>3</sub> precursor concentrations. It may be due to the higher content of WO<sub>3</sub> precursor and thicker film formation. Figure 3 (b) shows the plot of  $(\alpha hv)^{1/2}$  versus  $hv$  graph. The optical band gap energy of WO<sub>3</sub> films was determined from Tauc's plot equation,  $(\alpha hv)^n = A (hv - E_g)$ , where  $\alpha$  is absorption coefficient,  $hv$  is incident light energy, and  $E_g$  is the band gap of the material. The exponent 'n' indicates the transition type of the material. The value of n is 1/2 for indirect transition and 2 for direct transition. The value of the optical band gap energy of WO<sub>3</sub> films was obtained by extrapolating the linear portion of the curve to photon energy axis. The optical band gap energy of WO<sub>3</sub> films decreased from 3.3 eV to 2.8 eV when the WO<sub>3</sub> precursor increased from 0.03 M to 0.07 M. The decreasing trend of optical transmission and band gap energy was observed upon increasing WO<sub>3</sub> precursor concentrations. The band gap energy depends on the materials preparation conditions.



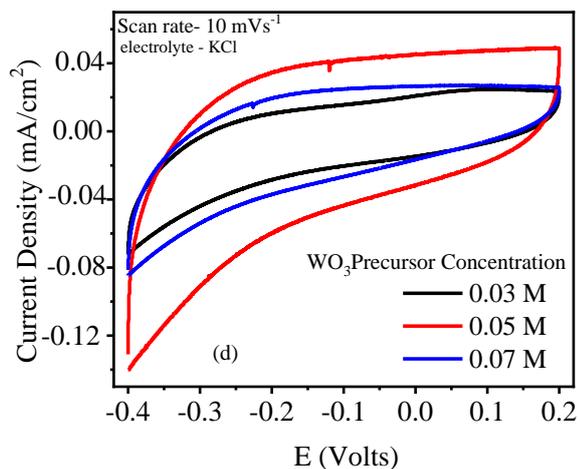
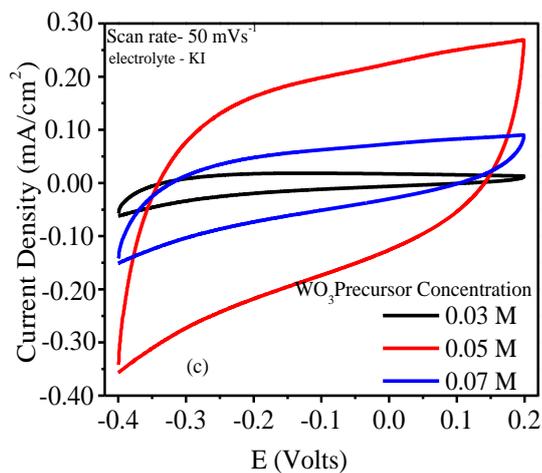
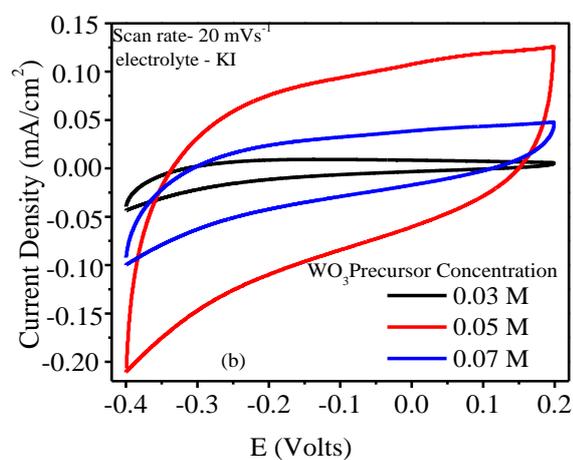
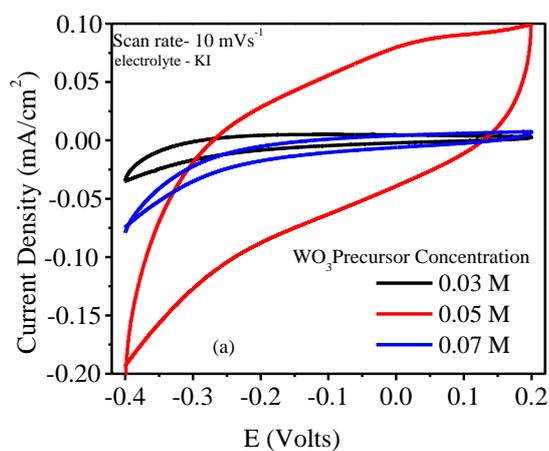
**Figure 3** (a) Optical transmission spectra and (b) Plot of  $(\alpha hv)^{1/2}$  versus  $hv$  of WO<sub>3</sub> films prepared from different WO<sub>3</sub> precursor concentrations

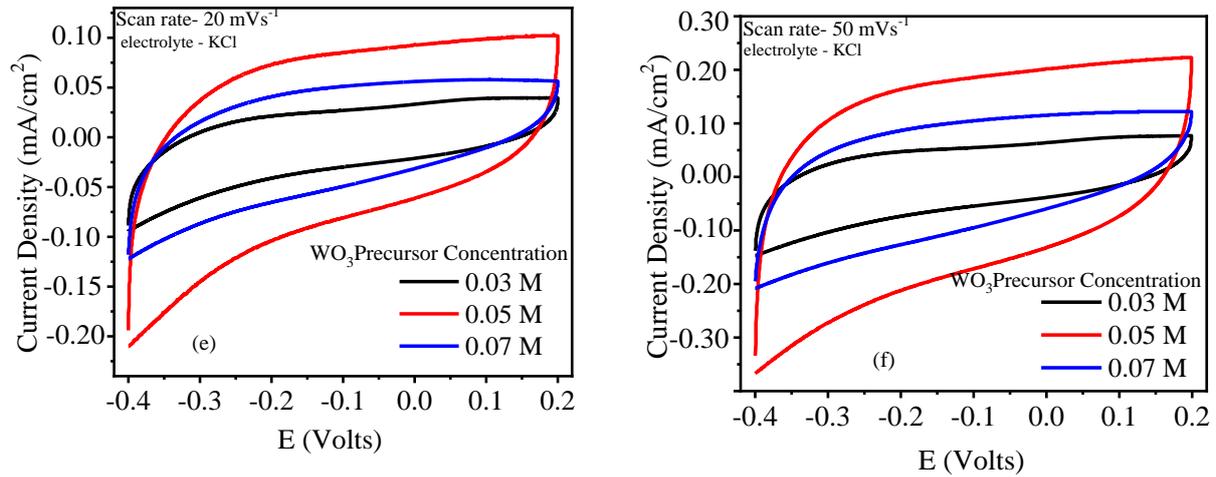
### Electrochemical Properties of WO<sub>3</sub> Films

The electrochromic behavior of WO<sub>3</sub> films was tested by cyclic voltammetry (CV) for different WO<sub>3</sub> precursor concentrations. Figure 4 (a-f) shows the CV graph of WO<sub>3</sub> films prepared from various WO<sub>3</sub> precursor concentrations in KI and KCl electrolyte for each scan rate. The cyclic voltammograms of WO<sub>3</sub> films were recorded in the potential range from -0.4 V to +0.2 V for each scan rate. From the CV measurement, the values of the anodic peak current densities of WO<sub>3</sub> films are extracted. The obtained values of the anodic peak current density of WO<sub>3</sub> films are listed in Table 1. The highest anodic peak current density of WO<sub>3</sub> films was observed at 0.05 M for each scan rate. The highest anodic peak current density of WO<sub>3</sub> films was observed at WO<sub>3</sub> precursor concentration of 0.05 M. The values of the anodic peak current density of WO<sub>3</sub> films with different WO<sub>3</sub> precursor concentrations are varied upon changing electrolyte. It would be related to the surface roughness of the WO<sub>3</sub> films. Since, the ion insertion/extraction reaction depends on the number of reaction sites on the surface.

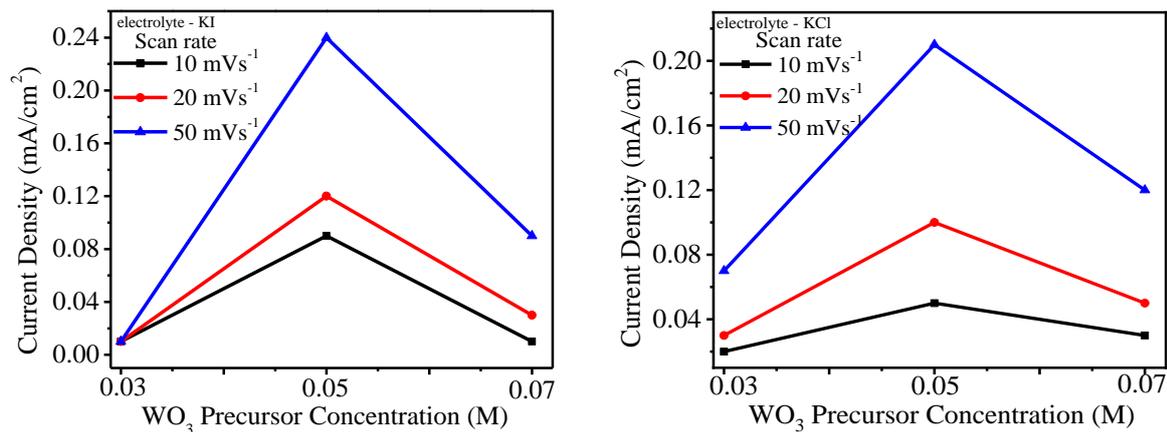
**Table 1** The value of anodic peak current densities of WO<sub>3</sub> films with varying WO<sub>3</sub> precursor concentrations in KI & KCl electrolyte

Scan Rate (mV/s)	WO <sub>3</sub> precursor concentrations	Potential (V)	Anodic Peak Current Density (mA/cm <sup>2</sup> ) KI electrolyte	Anodic Peak Current Density (mA/cm <sup>2</sup> ) KCl electrolyte
10	0.03 M	0.08	0.01	0.02
	0.05M	0.08	0.09	0.05
	0.07 M	0.08	0.01	0.03
20	0.03 M	0.08	0.01	0.03
	0.05M	0.08	0.12	0.10
	0.07 M	0.08	0.03	0.05
50	0.03 M	0.08	0.01	0.07
	0.05M	0.08	0.24	0.21
	0.07 M	0.08	0.09	0.12





**Figure 4** (a–f) Cyclic voltammogram of WO<sub>3</sub> films varying WO<sub>3</sub> precursor concentrations (0.03 M, 0.05 M and 0.07 M) in KI & KCl electrolytes



**Figure 5** Plot of current density (mA/cm<sup>2</sup>) vs WO<sub>3</sub> precursor concentration (M) in KI and KCl electrolyte

**Coloration efficiency and Diffusion coefficient of WO<sub>3</sub> Films**

The electrochromic behaviors specially coloration efficiency of WO<sub>3</sub> films in KI and KCl electrolyte were studied. Cyclic voltammetry was used to obtain the electrochemical properties of the WO<sub>3</sub> films under oxidation state. The value of peak current density gave a rough estimation of electrochromical activity of the working electrode. The main parameter that will be determined from the cyclic voltammetry was the current peak *I<sub>p</sub>*, which depends strongly on the diffusion coefficient. Anodic peak refers here to the current peak caused by insertion of K<sup>+</sup> ions into the WO<sub>3</sub> films, while the cathodic peak refers to exertion of K<sup>+</sup> ions. The WO<sub>3</sub> films have electrochromic properties, which is associated with the electrochemical intercalation and de-intercalation of K<sup>+</sup> ions and electrons into the WO<sub>3</sub> film. The experiments were recorded for different scan rates like 10 mVs<sup>-1</sup>, 20 mVs<sup>-1</sup> and 50 mVs<sup>-1</sup> in the KI and KCl electrolyte. During coloration the WO<sub>3</sub> film becomes blue color and turns into colorless during the bleaching. The diffusion coefficients of K<sup>+</sup> ions during intercalation and de-intercalation can be calculated by employing the Randles-Servick equation [P.R.Patil *et al.*, (2004) “Preparation and Characterization of Spray Deposited n-type WO<sub>3</sub> thin Films for Electrochromic Devices” Materials Research Bulletin, United States, vol.39, pp.1479-1489].

$$i_p = 2.72 \times 10^5 n^{3/2} D^{1/2} C v^{1/2} \tag{1}$$

where D is the diffusion coefficient of K<sup>+</sup> ions; v, scan rate; C, concentration of electrolyte solution; n, number of electrons and it is assumed to be 1 and *i<sub>p</sub>* is the peak current density. Figure 4 (a-f)

show the CV curves recorded for WO<sub>3</sub> films (0.05 M, 400°C) with different scan rate in potassium iodide (KI) and potassium chloride (KCl) electrolytes. Table 2 shows the various electrochemical parameters: the scan rate and diffusion coefficient extracted from Figure 6 (a-c). In the present work, the calculated diffusion coefficient is  $(1.08 - 1.35) \times 10^{-12} \text{ cm}^2\text{s}^{-1}$  in KI electrolyte and  $(1.09 - 1.38) \times 10^{-12} \text{ cm}^2\text{s}^{-1}$  in KCl electrolyte. Diffusion coefficient values for the K<sup>+</sup> ion process using different concentrations of KI, KCl electrolyte were of the order of  $(1.35 - 3.10) \times 10^{-12} \text{ cm}^2\text{s}^{-1}$  for spray pyrolyzed. From the electrochemical analysis it is observed that the films have changed their colors in accordance with the applied potential. Also the films have the capability to withstand in the electrolyte throughout the entire scan rates; it confirms the better reversibility of the films in the electrochromic process. The electrolyte content and the porosity play a great role in the diffusion coefficient by increasing the diffusion coefficient value with the K<sup>+</sup> ions. From this, WO<sub>3</sub> films prepared in the present work would be suitable for development of low cost electrochromic cells.

Figure 7 shows the UV-Vis spectra of the WO<sub>3</sub> films at colored and bleached state upon being subjected to -0.4 and 0.2 V in KI and KCl solution for 240 s. As shown in Figure 7, both films show a high transmittance which is as high as 80% at positive voltage. When applied to a negative voltage, a significant absorbance in the wavelength ranging from 550 nm to near IR region caused by the K<sup>+</sup> intercalation is observed. Both films turned from blue at colored state to colorless at bleached states as the voltage switched from negative to positive. The coloration efficiency,  $\eta$ , was calculated at constant  $\lambda = 550 \text{ nm}$  by applying a constant current on it ( $I = 1 \text{ mA}$ ) in the 0.1 M KI and KCl electrolyte solutions for 240 seconds. The charge,  $Q$ , inserted in the sample of (area  $A = 2 \text{ cm}^2$ ) is:

$$Q = I \times t \quad (2)$$

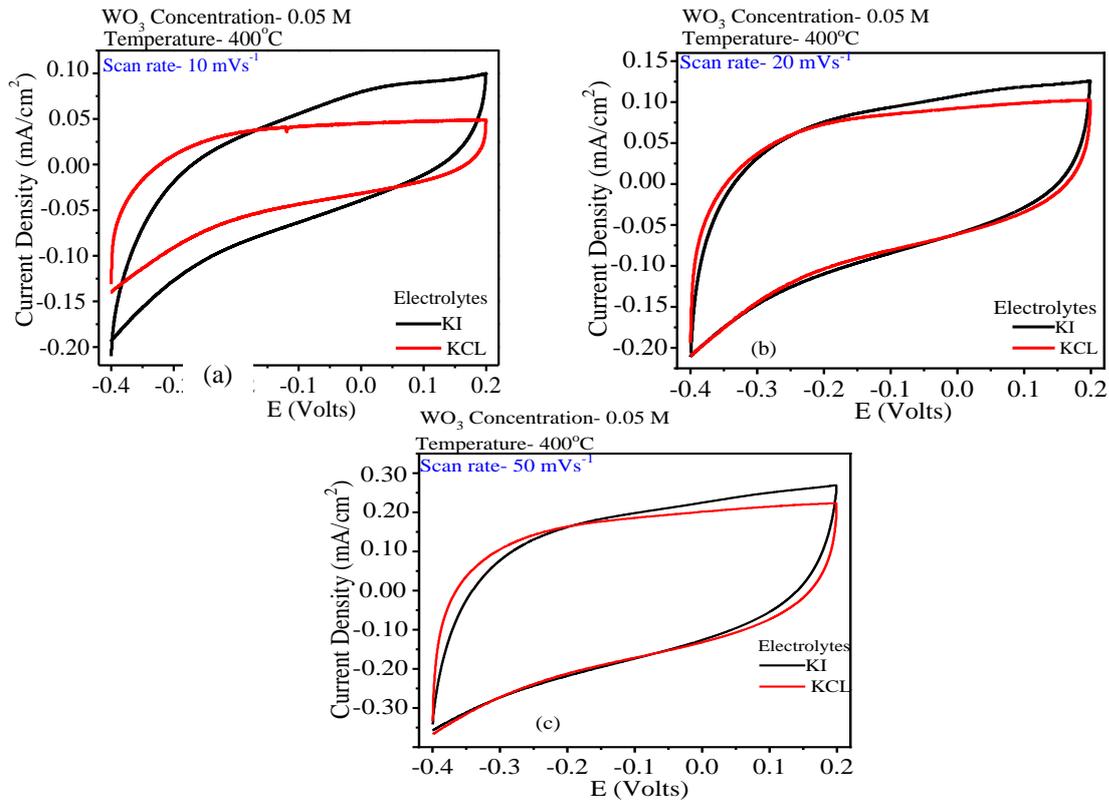
$$= 0.001 \times 240 = 0.24 \text{ C},$$

$$Q/A = 0.24/4 = 0.06 \text{ Ccm}^{-2}. \quad (3)$$

By using the equation (3),  $\eta$  can be calculated as follows:

$$\eta = \frac{\Delta OD}{\Delta Q} = \frac{\log\left(\frac{T_b}{T_c}\right)}{\left(\frac{Q}{A}\right)}, \quad (4)$$

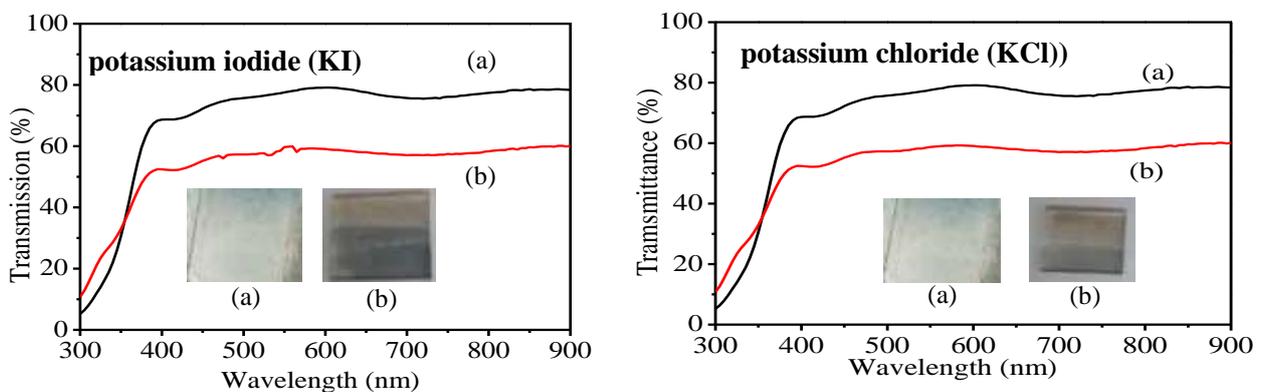
where  $\Delta OD$  is the optical density difference,  $T_b$  and  $T_c$  are transmittance at bleached state and colored state respectively as shown in Figure 6. The calculated coloration efficiency is  $1.89 \text{ cm}^2\text{C}^{-1}$  in KI and  $2.01 \text{ cm}^2\text{C}^{-1}$  in KCl electrolyte. The color change is believed to be associated with the intercalation (deintercalation) of the K<sup>+</sup> ions into (or out from) the WO<sub>3</sub> films. The the diffusion coefficient of WO<sub>3</sub> films in KI and KCl electrolyte solutions are not significantly varied. Consequently, the coloration efficiencies (color change) are not significantly varied in KI and KCl electrolytes.



**Figure 6 (a–c)** Cyclic voltammograms of  $WO_3$  films ( $400^\circ C$  and  $0.05M$ ) in KI and KCl electrolyte

**Table 2** Diffusion coefficients of potassium ions ( $K^+$ ) calculated using Randles-Sevcik equation for  $WO_3$  films in KI and KCl solution

Scan Rate (mV/s)	Diffusion Coefficient (anodic peak) ( $\times 10^{-12} cm^2 s^{-1}$ )	
	Electrolyte (KI)	Electrolyte (KCl)
10	1.35	1.38
20	1.14	1.14
50	1.08	1.09



**Figure 7** Transmittance spectra of  $WO_3$  films with potassium iodide (KI) and potassium chloride (KCl) at (a) bleached state and (b) colored state

## Summary and Conclusion

Tungsten trioxide (WO<sub>3</sub>) films were fabricated by spray pyrolysis method varying WO<sub>3</sub> precursor concentrations. The XRD measurement indicated that WO<sub>3</sub> film annealed at 300 °C shows the amorphous phase. When the annealing temperatures increased to 400 °C and 500 °C, WO<sub>3</sub> films reveal the orthorhombic structure. The main diffraction peak positions of WO<sub>3</sub> films annealed at 400 °C and 500 °C remain unchanged but the peak intensities declined. The stronger peak intensity of WO<sub>3</sub> films was observed at annealing temperature 400 °C. The optical transmission and band gap energy of WO<sub>3</sub> films decreased upon increasing WO<sub>3</sub> precursor concentrations. It may be due to the higher content of WO<sub>3</sub> precursor and thicker film formation. In the case of change in WO<sub>3</sub> precursor, the highest anodic peak current density was observed in WO<sub>3</sub> films (0.05 M) at each scan rate. Annealing temperature (400°C) and WO<sub>3</sub> precursor concentration (0.05 M) were selected for determining the performance of electrochromic device. The calculated diffusion coefficient is  $1.08 - 1.35 \times 10^{-12} \text{ cm}^2\text{s}^{-1}$  in KI electrolyte and  $1.09 - 1.38 \times 10^{-12} \text{ cm}^2\text{s}^{-1}$  in KCl electrolyte. From the electrochemical analysis it is observed that the films have changed their colors in accordance with the applied potential. Also the films have the capability to withstand in the electrolyte throughout the entire scan rates; indicating the better reversibility of the films in the electrochromic process. The higher diffusion coefficient indicates a larger contact area and greater porosity resulting in faster ion insertion/extraction. The coloration efficiency,  $\eta$  calculated at constant  $\lambda = 550 \text{ nm}$  is found to be  $1.89 \text{ cm}^2\text{C}^{-1}$  in KI and  $2.01 \text{ cm}^2\text{C}^{-1}$  in KCl electrolyte.

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